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CVD growth of perovskite/graphene films for high-performance flexible image sensor

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ABSTRACT

Hybrid perovskite possesses excellent photoelectric properties, including large light-absorption capacity and high carrier mobility, and is an ideal light-absorbing material for photoelectric devices. The grain size and compactness of hybrid perovskite are key factors affecting the performance of photoelectric devices. The photocurrent and photoresponsivity of these devices are relatively low because of the rapidly recombined photoexcited electron-hole pairs in hybrid perovskite. Herein, we develop a facile two-step chemical vapor deposition (CVD) method to synthesize a high-quality van der Waals (vdWs) MAPbI₃/graphene heterostructure for high-performance image sensor. We introduced inorganic sources (PbI₂) to vdWs epitaxially grown PbI₂ film on a seamless graphene monolayer film template through CVD. Methylammonium iodide (MAI) was then reintroduced to prepare the vdWs MAPbI₃/graphene heterostructure. The MAPbI₃ layer is composed of densely packed, large-size grains and displays a smooth surface. High photoresponsivity of 10⁷ A/W is achieved in the corresponding photodetector. Inspired by the human visual system, we designed a flexible photodetector array containing (24 × 24) pixels, achieving perfect image recognition and color discrimination. Our study may greatly facilitate the construction of high-performance optoelectronic devices in artificial retina, biomedical imaging, remote sensing, and optical communication.

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1. Introduction

Photodetector, which converts light into electrical signals has been widely used in industries and science, including sensing and communication [1–4]. Its key component is the light-absorbing material. In particular, hybrid perovskite (e.g., methylammonium halide lead (MAPbX₃, X = Cl, Br, I)) has recently received extensive attention [5–9] and possesses excellent photoelectric properties, including large light-absorption capacity, high carrier mobility, suitable band gap and long electron-hole diffusion length, making it an ideal light-absorbing material for photodetector [10–13]. However, the performance of a hybrid perovskite-based photodetector highly depends on the quality of the hybrid

perovskite material, especially its grain size and film compactness. In addition, the photoexcited electron-hole pairs in the hybrid perovskite recombine within a few picoseconds (i.e., within the lifetime of photoexcited electrons), thereby reducing photocurrent and photoresponsivity [5,14–16]. Therefore, preparing high-quality perovskite and improving the deficiency of the hybrid perovskite itself are necessary to obtain a high-performance photodetector.

Van der Waals (vdWs) heterostructure, which is prepared through the construction of materials and interfaces, can be used to improve light-substance interactions to build high-performance photoelectric devices [17–19]. Graphene has received great attention due to its unique properties [20], including high mobility, broadband absorption, and fast response time, thereby enabling its enormous application potential in photonics and optoelectronics [21–23]. When graphene and hybrid perovskite form a vdWs heterostructure, the empty state in the hybrid perovskite

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valence band produced through photon absorption will be filled by the electrons in the proximal graphene layer. Therefore, the recombination of photoexcited electron-hole pairs in the hybrid perovskite is reduced, and the electrons remain in the conduction band without being attenuated [5,14]. Furthermore, the graphene surface has no dangling bonds, in which sp^2 -hybridized carbon atoms are closely packed into a two-dimensional (2D) honeycomb lattice. Owing to the low energy barrier of epitaxial growth, graphene can be used as an ideal template to facilitate the vdWs epitaxial growth of 2D materials and consequently enable the synthesis of high-quality hybrid perovskite films [24–27]. Therefore, the presence of graphene can produce high-quality perovskite, improve the interface light-substance interaction, and boost the photocurrent and performance of photodetectors.

In this work, we developed a facile two-step CVD method to prepare a high-quality vdWs MAPbI₃/graphene heterostructure and verify its application for image sensors. The hybrid perovskite was synthesized on a seamless monolayer graphene film by sequentially introducing inorganic sources (PbI₂) and organic sources (methylammonium iodide, MAI) through CVD. The seamlessly large-area grain-boundary-free graphene film was grown by CVD and used as a growth template. The vdWs epitaxially grown hybrid perovskite layer is dense and smooth and possesses large-size grains. Basing on the vdWs MAPbI₃/graphene heterostructure, we constructed a photodetector and obtained an extremely high photoresponsivity of $\sim 10^7$ A/W and a low response time of 50 ms. Inspired by the human visual system, we also designed an integrated (24 × 24) pixels flexible photodetector array as an image sensor, in which the vdWs MAPbI₃/graphene heterostructure was used as cone cells. We demonstrated the photon detection, resolution of the color and clear shape imaging of the flexible image sensor. The high-quality vdWs MAPbI₃/graphene heterostructure has extensive application potentials in integrated optoelectronic circuits, biomedical imaging, and optical communication in the future.

2. Materials and methods

2.1. Synthesis of high-quality seamless monolayer graphene films

The monolayer graphene film was synthesized on copper (Cu) foil (25 μ m thick, Alfa Aesar Inc.) by CVD. The Cu foil was cleaned in hot acetic acid for 30 min to remove the surface oxides and impurity, and then washed with acetone, isopropanol and deionized (DI) water for 10 min, respectively. The cleaned Cu foil was dried with a nitrogen gun. We placed the Cu foil into the center of the furnace and then heated to 1070 °C with a flow of hydrogen (H₂, 100 sccm (standard cubic centimeter per minutes), 99.999% purity) and Ar (500 sccm, 99.999% purity) in 50 min. This system was kept for 5 h and then methane (CH₄) (0.2 sccm, 99.999% purity) was introduced. The growth was terminated after 50 min by stopping the CH₄ flow and then cooled down the system in Ar and H₂.

2.2. Transfer of graphene

The transfer process was as follows: (1) Spin coating 300 nm polymethyl methacrylate (PMMA, 3000 r/min for 30 s) on the graphene/Cu surface; (2) bake the sample in thermal drying oven at 170 °C for 10 min; (3) put the sample into (NH₄)₂S₂O₈ solution (0.5 mol/L) to etch the Cu foil; (4) pick up the floating PMMA/graphene film with a clean glass slide and wash in deionized water for 3 times; (5) dry in a blast oven for 30 min to remove water; (6) dip into acetone to dissolve the PMMA layer.

2.3. Synthesis of PbI₂ films on monolayer graphene film templates

PbI₂ powder (98%, J&K Chemicals) was used as the inorganic source and was put into a quartz tube. The graphene template substrate was put into the downstream region of the tube. The quartz tube was first evacuated to a pressure of 3 Pa and then Ar (200 sccm) was introduced. Ten minutes later, the furnace was heated to 320 °C and held for another 10 min. Finally, the furnace was cooled down naturally with flowing Ar to ambient temperature.

2.4. Synthesis of MAPbI₃ films on monolayer graphene film templates

CH₃NH₃I powder (99.999%, J&K Chemicals) was used as the organic source and was put into a quartz tube. The PbI₂/graphene substrate was put into the downstream region. The quartz tube was first evacuated to a pressure of 3 Pa and then Ar (200 sccm) was introduced. Ten minutes later, the furnace was heated to 180 °C and held for another 30 min. Finally, the furnace was cooled down naturally with flowing Ar to ambient temperature. The obtained sample was annealed at 100 °C in air for 50 min for crystal transformation.

2.5. Fabrication of flexible photodetector array

The fabrication process was as follows: (1) Spin coat a layer of NR9-3000PY type negative photoresist on cleaned PET film. After photoetching, deposit 50 nm Au as the X-direction electrode. (2) Spin coat a layer of AZ601 type positive photoresist on the sample of (1). After photoetching, deposit 100 nm alumina as the insulating layer. (3) Spin coat a layer of AZ601 type positive photoresist on the sample of (2). After photoetching, deposit 50 nm Pt as the Y-direction electrode. (4) Transfer a layer of graphene on the sample of (3) and then spin coat a layer of AZ601 type positive photoresist. After photoetching, etch graphene. (5) Spin coat a layer of AZ601 type positive photoresist on the sample of (4) and then grow PbI₂ crystals in a CVD system. (6) Remove the photoresist with acetone and left PbI₂/graphene film on the substrate. (7) Synthesis MAPbI₃ perovskite by introducing CH₃NH₃I. (8) Remove the excess CH₃NH₃I with isopropanol and then obtain the flexible photodetector array.

3. Results and discussion

3.1. Fabrication and characterization of the PbI₂ grown on monolayer graphene templates

Fig. 1a illustrates a schematic diagram of the preparation of high-quality vdWs MAPbI₃/graphene heterostructure on a graphene template (see details in Materials and methods). The graphene template is a seamless monolayer graphene film composed of aligned millimeter graphene domains grown on copper (Cu) foil by CVD (**Fig. S1** online). The film has few cracks or defects, a dangling-bond-free surface, and a 2D honeycomb lattice. This film is the key to achieving the epitaxial growth of high-quality perovskite films. First, we introduced inorganic sources (PbI₂) to synthesize high-quality PbI₂ on graphene films. The organic sources (MAI) were reintroduced to prepare the vdWs MAPbI₃/graphene heterostructure according to the synthesized vdWs PbI₂/graphene heterostructure.

Fig. 1b and c show the top and side views of the lattice structure of the PbI₂/graphene heterostructure, respectively. PbI₂ has a layered crystal structure, and it is mainly the vdWs force between the PbI₂ and the graphene layer. The orientation of PbI₂ crystals is derived from the lattice-matching between graphene and PbI₂ and the relatively low total energy of the entire heterostructure.

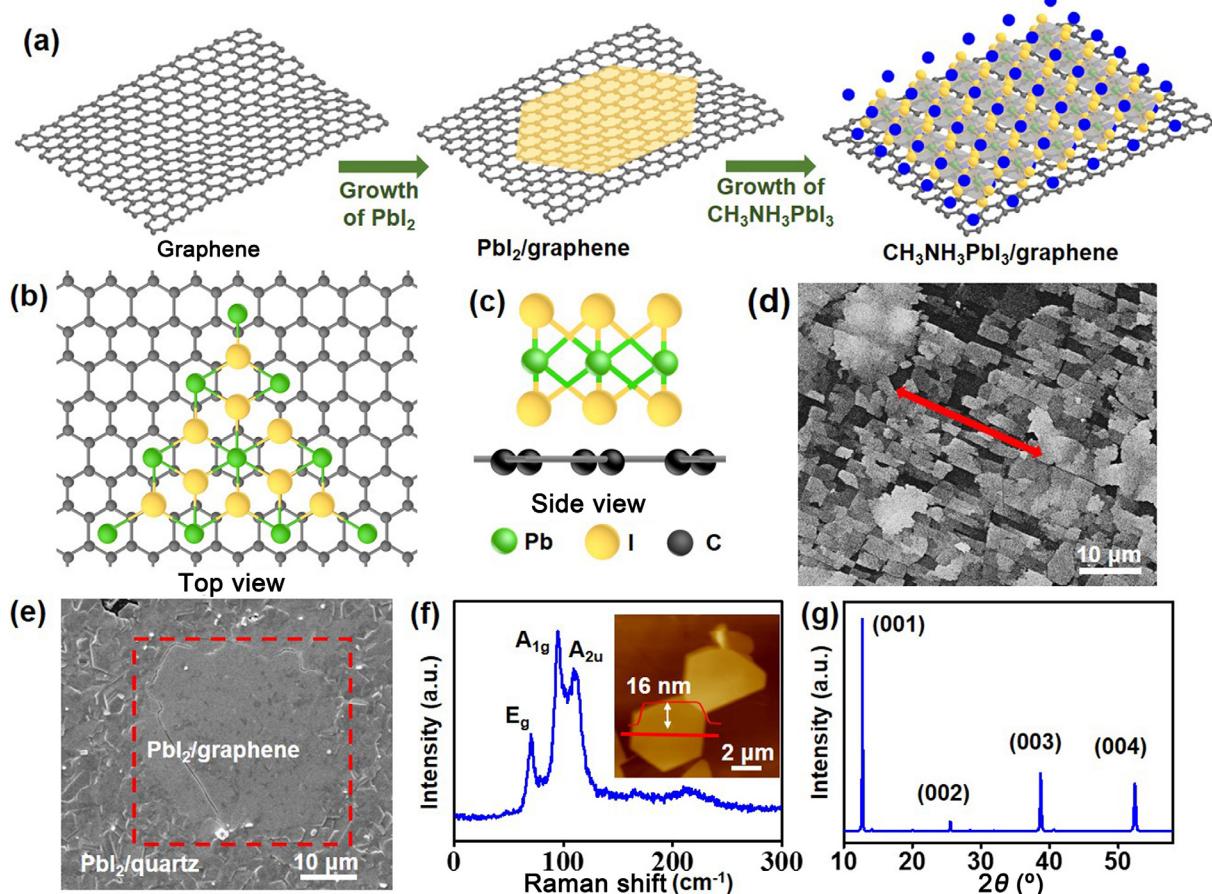


Fig. 1. (Color online) Illustration of the preparation process of vdWs high-quality MAPbI₃/graphene heterostructure and the structure of the PbI₂/graphene heterostructure. (a) Schematic diagram showing the preparation process of vdWs MAPbI₃/graphene heterostructure by sequentially introducing PbI₂ and MAI through a two-step CVD. Schematic diagram showing the lattice structure of vdWs PbI₂/graphene heterostructure in top view (b) and side view (c). (d) SEM image of incompletely filmed PbI₂ sample grown on graphene. (e) SEM image of PbI₂ film grown on a single crystal graphene domain on a quartz substrate. Raman spectrum (f) and XRD profile (g) of a PbI₂ film grown on graphene. The inset picture in (f) is an AFM topography image of PbI₂.

system [28]. Fig. 1d is a scanning electron microscope (SEM) image of the PbI₂ crystals grown on graphene showing the aligned PbI₂ crystals, indicating the potential for seamless merging. Short growth time was intentionally used for this sample to clearly show the aligned PbI₂ crystals grown on the graphene template. Fig. 1e shows a SEM image of the PbI₂ crystals grown on a single crystal graphene domain on a quartz substrate. The PbI₂ films grown on graphene films are smoother than those grown on quartz, indicating the excellent epitaxial promotion effect of the graphene templates (see details in Fig. S2 online). Furthermore, we used transmission electron microscopy (TEM), atomic force microscopy (AFM), Raman spectroscopy, and X-ray diffraction (XRD) to systematically study the structure and chemical properties of the epitaxially grown vdWs PbI₂/graphene heterostructure. Fig. 1f shows the typical Raman spectrum of the PbI₂ sample revealing three characteristic peaks, namely, E_g (75 cm⁻¹), A_{1g} (96 cm⁻¹), and A_{2u} (113 cm⁻¹). The inset picture is an AFM topography image of PbI₂ domains grown on graphene, indicating a height of ~16 nm of the PbI₂ domain. The XRD pattern of the grown PbI₂ film shows (0 0 1), (0 0 2), (0 0 3), and (0 0 4) peaks occurring on the PbI₂ film, suggesting that 2D PbI₂ domains possess the 2H hexagonal crystalline structure (Fig. 1g). A low magnification TEM image of a PbI₂ domain shows a smooth surface, and the selected area electron diffraction pattern of the PbI₂ further confirms its single-crystallinity property (Fig. S3 online).

3.2. Characterization of vdWs MAPbI₃/graphene heterostructures

High-quality vdWs MAPbI₃/graphene heterostructure films can be grown on vdWs epitaxially grown PbI₂ films. Fig. 2a is a SEM image of a vdWs MAPbI₃/graphene heterostructure showing that the film is dense, and the size of the MAPbI₃ crystal grains is ~2 μm. The AFM image further proves the compactness and large grain size of the film (Fig. 2b). We intentionally synthesized the MAPbI₃ film on quartz for comparison to verify the high-quality MAPbI₃ film grown on graphene template (see details in Materials and methods). Fig. 2c is a SEM image of a MAPbI₃ film grown directly on quartz through a two-step CVD showing that the MAPbI₃ film has an average grain size of ~150 nm. Fig. 2d shows an AFM topography image of a MAPbI₃ film grown on quartz, thereby further revealing its small grain size and porous film properties. Compared with the perovskite grown directly on the quartz substrate, the perovskite film prepared by using graphene as a template has larger crystal grains and a denser film structure, and the crystallinity of the perovskite is remarkably improved.

We used photoluminescence (PL) spectroscopy, XRD, and X-ray photoelectron spectroscopy (XPS) to further study the structure and chemical composition of the synthesized MAPbI₃ film. Fig. 2e is the typical PL spectra of the MAPbI₃ film grown on quartz and vdWs MAPbI₃/graphene heterostructure. It shows that the PL quenching in the vdWs MAPbI₃/graphene heterostructure was

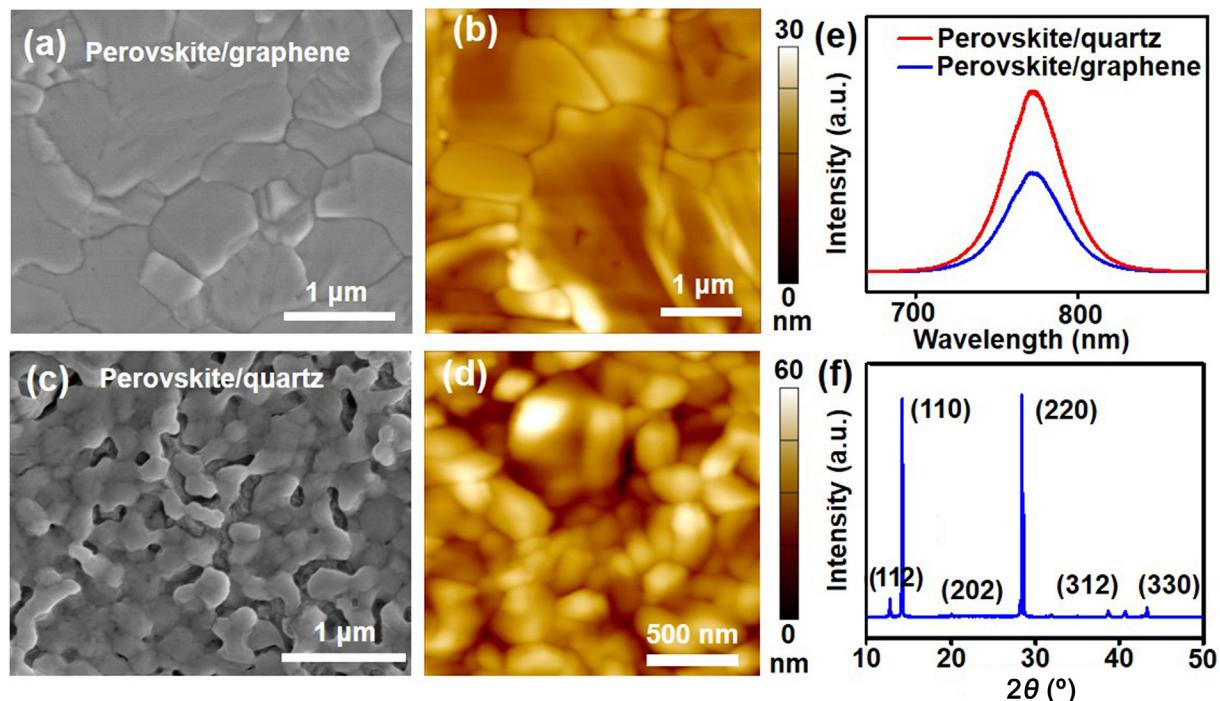


Fig. 2. (Color online) Characterization of MAPbI_3 grown on vdWs $\text{PbI}_2/\text{graphene}$ heterostructures and quartz. SEM (a) and AFM (b) topography images of a vdWs $\text{MAPbI}_3/\text{graphene}$ heterostructure, respectively. SEM (c) and AFM (d) topography images of a MAPbI_3 film grown on quartz, respectively. (e) Photoluminescence spectra of a MAPbI_3 film and a vdWs $\text{MAPbI}_3/\text{graphene}$ heterostructure. (f) XRD pattern of a vdWs $\text{MAPbI}_3/\text{graphene}$ heterostructure.

lower than that in the MAPbI_3 film. In the presence of graphene, the electrons in the graphene layer were transferred to the proximal hybrid perovskite layer to fill the empty state in the hybrid perovskite valence band produced by photon absorption, reducing the recombination of photoexcited electron-hole pairs in hybrid perovskite and thereby contributing to the high-performance of the photodetectors [29]. Fig. 2f shows the XRD pattern of the MAPbI_3 film grown on graphene. It shows diffraction peaks at 14.2° , 28.4° and 43.2° , which can be assigned to the $(1\ 1\ 0)$, $(2\ 2\ 0)$ and $(3\ 3\ 0)$ crystal faces of the tetragonal phase of MAPbI_3 and are consistent with previous results [1]. The relative intensity of the $(1\ 1\ 0)$ and $(2\ 2\ 0)$ peaks are very high, and the full width at half-maximum of the $(1\ 1\ 0)$ and $(2\ 2\ 0)$ peaks are only 0.024 and 0.016, respectively, indicating the high crystallinity of the as-grown MAPbI_3 film [30]. The XPS spectrum of the vdWs $\text{MAPbI}_3/\text{graphene}$ heterostructure exhibited four elements, namely, C, N, I, and Pb (Fig. S4 online). The UV-visible absorption spectra of the vdWs $\text{MAPbI}_3/\text{graphene}$ heterostructure showed its great light-absorption performance in the visible range (Fig. S5 online). This finding indicates that the vdWs $\text{MAPbI}_3/\text{graphene}$ heterostructure can fully adsorb light energy and produce electron-hole pairs, thereby confirming its effective photodetection under visible light irradiation.

3.3. Photoelectric performance of the vdWs $\text{MAPbI}_3/\text{graphene}$ heterostructure based photodetector

We constructed photodetectors based on the high-quality vdWs $\text{MAPbI}_3/\text{graphene}$ heterostructure to study the photoelectric properties of the as-prepared heterostructures (Fig. 3). Fig. 3a shows the schematic configuration of the photodetector in which vdWs $\text{MAPbI}_3/\text{graphene}$ heterostructure is used as a light-absorbing layer. Fig. 3b shows the photocurrent of the photodetector with different incident illumination powers under 633 nm illumination wavelength at 1 V bias. The photocurrent was linearly proportional to the incident illumination power. Fig. 3c exhibits the

current-voltage ($I-V$) curves of the photodetector as a function of illumination wavelength with a fixed incident illumination power of $30 \mu\text{W}$. Good light responsiveness was observed. When the illumination wavelength was close to the infrared band, the degree of light response was lowered, which corresponds to the light-absorption range of MAPbI_3 . Fig. 3d shows the illumination power-dependent $I-V$ curves of the photodetector under an illumination wavelength of 633 nm. The photocurrent of the device increased with the illumination power and voltage.

Photoresponsivity describes the photoelectric conversion capability of the device and is an important indicator for photodetectors that is related to the device material, illumination power, and wavelength of light. Photoresponsivity (R) can be defined as $R = (I_{\text{light}} - I_{\text{dark}})/(P_0 \times S)$, where I_{light} is the photocurrent when the light is ON, I_{dark} is the dark current when the light is OFF, P_0 is the irradiance power density, and S is the effective illuminated area. Fig. 3e shows the relationship between photoresponsivity and illumination power within a fixed wavelength of 633 nm. At 0.1 V, the photodetector exhibited an impressive photoresponsivity value of $\sim 2700 \text{ A/W}$ with an illumination power of 100 nW . We also prepared a photodetector based on a MAPbI_3 film without graphene for comparison (Fig. S6 online). The MAPbI_3 photodetectors showed a photoresponsivity value of $\sim 0.2 \text{ A/W}$ with an illumination power of 100 nW , which is significantly lower than that of the $\text{MAPbI}_3/\text{graphene}$ based photodetector. It shows that a high photoresponsivity can be achieved with weak light intensity. A reduced light intensity lowers the carrier concentration in the sensitizer layer, which generally reduces the electron-hole recombination. An almost linear relation was observed between photoresponsivity and illumination power ($R \propto P^{-1}$) (Fig. S7 online). When the power was reduced to 1 pw, the photoresponsivity was $\sim 10^7 \text{ A/W}$. This value is comparable or higher than that for previous $\text{MAPbI}_3/\text{graphene}$ film-based photodetectors [1,5,16,29,31-39].

Furthermore, we studied the periodic response capability and flexibility of the proposed device. The photo-switching characteristics of

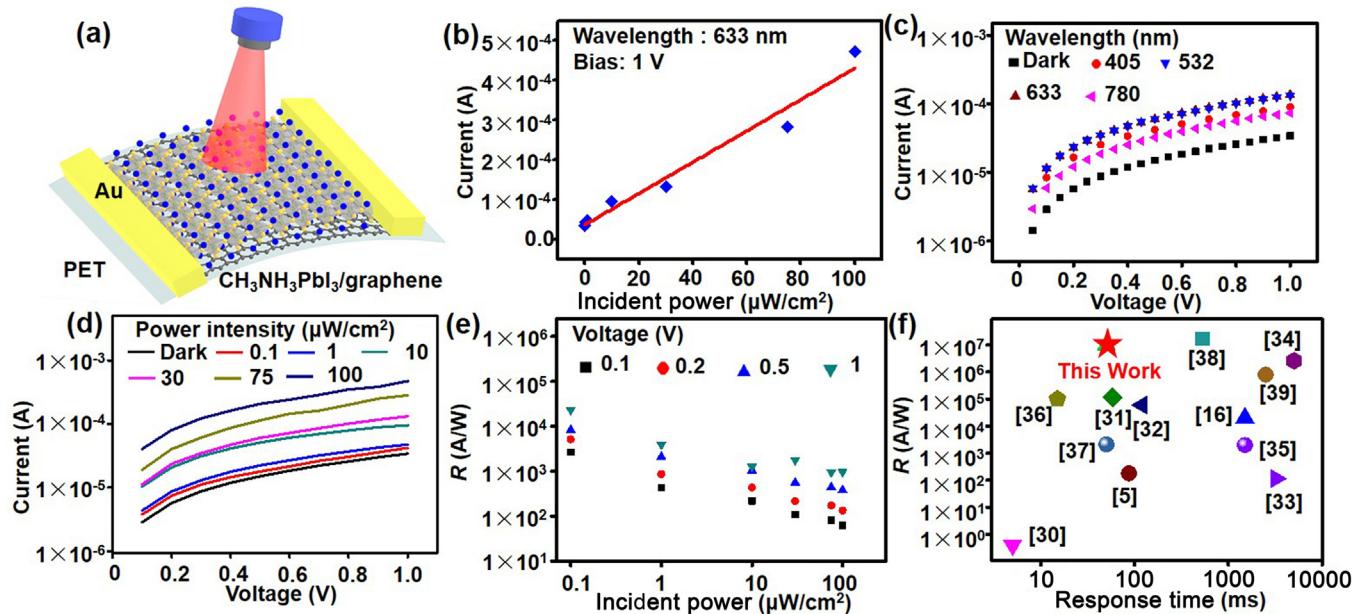


Fig. 3. (Color online) Schematic configuration and performance of the vdWs MAPbI₃/graphene heterostructure based photodetector. (a) Schematic diagram of the photodetector. (b) Photocurrent vs. incident illumination power under an illumination wavelength of 633 nm at 1 V bias. (c) I-V curves of the photodetector under various illumination wavelength. The incident illumination power is 30 μW . (d) I-V curves of the photodetector under various illumination power. The illumination wavelength is 633 nm. (e) Photoresponsivity of the photodetector under different voltages and illumination power. (f) Comparison of the photoresponsivity and response time of our vdWs MAPbI₃/graphene heterostructure based photodetector with the previous reported MAPbI₃/graphene film based photodetector.

the photodetector were measured under alternating light illumination (30 μW , 633 nm) with the gate and drain voltages of 0 and 0.1 V, respectively (Fig. S8 online). The device shows a stable performance with cyclic photocurrents that follow light signals. Response time is defined as the time used by the photocurrent to increase from

null to 80% of the saturation level [40]. The response time was less than 50 ms, thereby indicating the rapid response of the device (Fig. S9 online). We also used a polyimide (PI) film as a flexible substrate to prepare a MAPbI₃/graphene heterostructure based flexible photodetector, which showed no significant performance

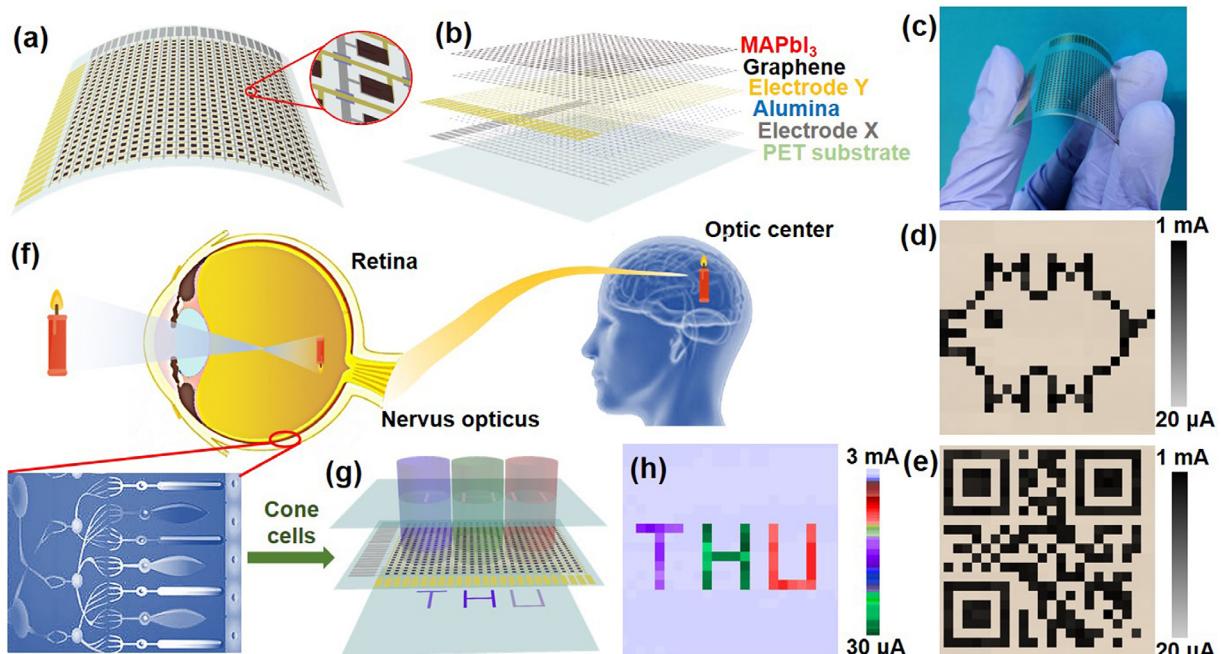


Fig. 4. (Color online) Preparation of the flexible image sensor based on vdWs MAPbI₃/graphene heterostructure and its application in image recognition. (a, b) Schematic illustration of the as-fabricated flexible (24 \times 24) pixels image sensor on PET substrate. (c) Photograph of a flexible image sensor. (d, e) The corresponding output images of a cartoon pig in (d) and QR code of "photodetector" in (e) under an light illumination (30 μW , 633 nm). (f) Schematic illustration of the multicolor imaging systems in human visual system. (g) Schematic diagram of our image sensor showing the vdWs MAPbI₃/graphene heterostructure acted as cone cells. (h) The corresponding output image of the flexible image sensor under different light illumination (75 μW , 380, 633 and 750 nm).

degradation after repetitive bending at 200 cycles. This finding indicates the excellent operational stability of the vdWs MAPbI₃/graphene heterostructure during repetitive bending tests (Fig. S10 online).

3.4. Preparation and characterization of flexible image sensors based on the vdWs MAPbI₃/graphene heterostructure

We also constructed an integrated flexible photodetector array, in which the vdWs MAPbI₃/graphene heterostructure acted as cone cells [41]. Fig. 4a shows a schematic diagram of the as-prepared flexible photodetector array with (24 × 24) pixels. The size of one pixel is 400 μm × 400 μm. Fig. 4b exhibits a cross-sectional view of the device. The photodetector array consists of six layers, in which polyethylene terephthalate (PET) was used as a flexible substrate. Between the Au (X-direction) and Pt (Y-direction) electrodes, an alumina layer was deposited as an insulating layer to prevent short-circuit. Then, a graphene film was transferred on to the electrode array and patterned, followed by the in-situ growing of MAPbI₃ perovskite. The vdWs MAPbI₃/graphene heterostructures acted as the light-sensitive materials (see in Materials and method and Figs. S11, S12 (online)). The MAPbI₃/graphene of each pixel is almost uniform and can work consistently. The data from all the pixels can be acquired simultaneously by the data acquisition module. Fig. 4c shows the as-prepared flexible photodetector array. We processed two photomasks, including a cartoon pig and QuickMark (QR) code of the word “photodetector” (Fig. 4d and e, respectively) under the illumination wavelength of 633 nm to reveal the imaging capabilities of the device. Excellent image recognition capabilities were observed.

Inspired by the human visual system, we demonstrated the potential of our devices in artificial retina. The human eye has two basic functions: color and shape recognition. We used different illumination wavelengths to examine the color recognition capabilities of the as-prepared device. Fig. 4f shows schematic of the imaging process from the human eye to the visual center of the human brain, where the cone cells in the retina identify the color of the object. Fig. 4g shows the schematic of our image sensor, which is imaged by filtering using three wavelength filters (380, 633, and 750 nm) under white light illumination. The results display that the photocurrent was the largest under 633 nm illumination and smallest under 750 nm, showing great potential for application in artificial visions.

4. Conclusion

In summary, we develop a facile two-step CVD method to synthesize high-quality vdWs MAPbI₃/graphene heterostructures by sequentially introducing PbI₂ and MAI with seamless monolayer graphene films as templates. The MAPbI₃ layer is composed of ultra-large grain-size MAPbI₃ domains and thus exhibits a smooth and dense film. The large-area grain-boundary-free graphene film is used as a template to facilitate the vdWs epitaxial growth of high-quality MAPbI₃ film. In the presence of graphene, electrons in the graphene layer are transferred to the proximal hybrid perovskite layer to fill the empty state in the hybrid perovskite valence band produced through photon absorption, thereby reducing the recombination of photoexcited electron-hole pairs in hybrid perovskite. This novel photodetector based on the vdWs MAPbI₃/graphene heterostructure shows high photoresponsivity (~10⁷ A/W), low operating voltage (0.1 V), and low response time (less than 50 ms). Inspired by the human visual system, we demonstrated the potential of our device in artificial retina, in which the vdWs MAPbI₃/graphene heterostructures acted as “cone cells”. Our study may open up an avenue to exploit the potential application

of high-quality hybrid perovskite heterostructures in optoelectronics, wearable devices, and artificial retina.

Declaration of Competing Interest

The authors declare that they have no conflict of interest.

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Author contributions

Yingying Zhang and Kailun Xia conceived the idea. Kailun Xia performed most of the experiments. Kailun Xia and Yingying Zhang prepared the draft. Wenqiang Wu, Mengjia Zhu, Zhe Yin, Haomin Wang, Shuo Li, Mingchao Zhang, Huimin Wang, and Haojie Lu participated in part of the experimental and discussed on the results. Caofeng Pan and Anlian Pan instructed the fabrication of the devices. All authors discussed the results and revised the manuscript.

Appendix A. Supplementary materials

Supplementary materials to this article can be found online at <https://doi.org/10.1016/j.scib.2019.12.015>.

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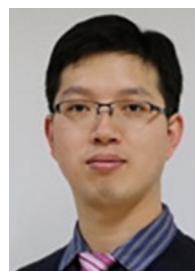
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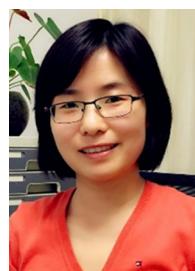
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